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# Scaling behaviour of the thermopower in a two-component composite near a percolation threshold

Ohad Levy and David J Bergman

School of Physics and Astronomy, Raymond and Beverly Sackler Faculty of Exact Sciences, Tel Aviv University, Tel Aviv 69978, Israel

Received 28 August 1991

Abstract. An exact relation between the bulk effective Seebeck coefficient  $\alpha_e$  of a composite conductor and the bulk effective electrical and thermal conductivities  $\sigma_e$  and  $\gamma_e$  is used to study the scaling behaviour of  $\alpha_e$  near percolation. The behaviour turns out to be quite rich, as a result of its dependence on three dimensionless small parameters, namely, the electrical and thermal conductivity ratios of the two components  $\sigma_I/\sigma_M$ ,  $\gamma_I/\gamma_M$ , and the distance away from the percolation threshold  $\Delta p \equiv p_M - p_e$ . The behaviour of the thermoelectric figure of merit  $Z_e$  in the different parts of the critical region is also discussed.

#### 1. Introduction

The thermoelectric effect is perhaps the simplest type of coupled field transport known in solids. Other types include magneto-transport, the study of which in percolating media has already yielded some interesting qualitative as well as quantitative results [1-4]. In this article we report on a theoretical study we have made of the critical behaviour of the thermoelectric properties of a two-component percolative medium. In order to do this, we exploit a field decoupling transformation which, when applied to the thermoelectric transport problem in a two-component composite, transforms it into a pair of uncoupled, simple conductivity-like transport problems in the same composite. This transformation, discovered by Straley [5], was already used by him to calculate the thermopower of a two-component composite. However the scaling behaviour near percolation is not discussed in that article. The results reported are largely numerical or graphical, and are confined to one particular category of components, which is the common one. Here we take up where [5] left off, and study other categories of components. We also use the scaling description of critical behaviour in simple, one-field conductivity, in order to work out explicitly the leading forms of the critical behaviour for the Seebeck coefficient in various sectors of the critical region. The fact that there are many such sectors with qualitatively different forms of behaviour is due to the appearance of three critical parameters in the problem, namely, the electrical conductivity ratio  $\sigma_I/\sigma_M$ , the thermal conductivity ratio  $\gamma_I/\gamma_M$  and the volume fraction difference  $\Delta p = p_M - p_c$ , where the subscripts M, I refer to the good and the bad conductor, respectively, and  $p_c$  is the good conductor percolation threshold. Because of this, even when all three parameters are small, there are six different asymptotic sectors in the critical region, corresponding to six different permutations or orderings by size of these parameters. The corresponding richness of different possible critical behaviours is reminiscent of the similar richness that was found, both

theoretically [3], and experimentally [4], in the critical behaviour of the low field Hall effect near a percolation threshold. In analogy with those results, we have found that in a good conductor-bad conductor mixture, the thermoelectric effect can, in some cases, be dominated by the good conductor even below its percolation threshold and, in some other cases, be dominated by the bad conductor even above that threshold.

The rest of this article is organized as follows. In section 2 we introduce our notation and summarize the theory of thermoelectricity in two-component composites, as well as the scaling description of simple, one-field conductivity near a percolation threshold. In section 3 we apply these elements to a discussion of the critical behaviour of the Seebeck coefficient and of the thermoelectric quality factor near a percolation threshold.

# 2. Theoretical framework

The electric current density that flows in a homogeneous material due to electric and temperature fields under linear response assumptions is given by

$$\boldsymbol{J}_E = -\boldsymbol{\sigma} \nabla \boldsymbol{\Phi} - \boldsymbol{\sigma} \boldsymbol{\alpha} \nabla \boldsymbol{T} \tag{1}$$

where  $\Phi$  is the electric potential, T is the temperature,  $\sigma$  is the electrical conductivity and  $\alpha$  is the thermoelectric coefficient or Seebeck coefficient. The entropy current density under the same conditions is given by

$$J_{s} = -\sigma \alpha \nabla \Phi - \frac{\gamma}{T} \nabla T \tag{2}$$

where  $\gamma$  is the thermal conductivity at zero electric field. These equations can be summarized in the following compact form

$$\mathbf{J} = \mathbf{S} \nabla \mathbf{\Phi} \tag{3}$$

where

$$\mathbf{J} \approx \begin{pmatrix} -J_E/e \\ -J_S/k \end{pmatrix} \qquad \Phi \equiv \begin{pmatrix} \nabla(e\Phi) \\ \nabla(kT) \end{pmatrix} \qquad \mathbf{S} \equiv \begin{pmatrix} \sigma/e^2 & \sigma\alpha/ek \\ \sigma\alpha/ek & \gamma/k^2T \end{pmatrix}.$$
(4)

The absolute charge of the electron e and Boltzmann's constant k have been included in these definitions in order to make both elements of  $\Phi$  have the same physical dimensions, and likewise for the elements of J and of the thermoelectric transport matrix **S**. The thermoelectric figure of merit is defined by

$$Z = \left(\frac{\gamma}{\sigma \alpha^2} - T\right)^{-1}.$$
 (5)

It determines the maximum efficiency of a thermoelectric heat pump constructed from this material [6].

Each component material of an inhomogeneous composite is characterized by its own transport matrix  $S_a$ . The material as a whole is then characterized by an effective transport matrix

$$\mathbf{S}_{e} = \begin{pmatrix} \sigma_{e}/e^{2} & \sigma_{e}\alpha_{e}/ek \\ \sigma_{e}\alpha_{e}/ek & \gamma_{e}/k^{2}T \end{pmatrix}$$
(6)

which relates the volume averaged fluxes  $\langle J \rangle$  to the volume averaged fields  $\langle \nabla \Phi \rangle$ 

$$\langle \mathbf{J} \rangle = \mathbf{S}_{\mathbf{c}} \langle \nabla \Phi \rangle. \tag{7}$$

Note that although we assume that the components are isotropic, we do not assume this for the composite: In the case of an anisotropic composite, our notation assumes that both of the volume averaged fields  $\langle \nabla \Phi \rangle$ ,  $\langle \nabla T \rangle$  point along the same direction, say along the x-axis, and that only the x-component of the volume averaged currents  $\langle J_E \rangle$ ,  $\langle J_S \rangle$  are determined by the 2×2 matrix  $\mathbf{S}_e$ .

In order to find useful forms for the bulk effective moduli  $\sigma_e$ ,  $\gamma_e$  and  $\alpha_e$  we use a diagonalization method similar to the one first suggested by Straley [5] and later generalized by Milgrom and Shtrikman [7]: We find a real symmetric matrix

$$\mathbf{d} = \begin{pmatrix} d_{11} & d_{12} \\ d_{12} & d_{22} \end{pmatrix}$$
(8)

such that  $dS_a d$  is diagonal for both components a = A, B. Explicit expressions can easily be found for the elements of the transformation matrix d in terms of the elements of  $S_A$  and  $S_B$  [8]. Having thus transformed the coupled-field transport problem into a pair of uncoupled, simple conductivity-like transport problems, we can describe the uncoupled transport in the composite medium by a bulk effective diagonal matrix  $S'_e$ 

$$\mathbf{S}_{e}^{\prime} = \begin{pmatrix} S_{e11}^{\prime} & 0\\ 0 & S_{e22}^{\prime} \end{pmatrix}.$$
(9)

The bulk effective matrix  $\mathbf{S}_e$  for the original, coupled field transport problem can now be obtained from  $\mathbf{S}'_e$  by the inverse transformation  $\mathbf{S}_e = \mathbf{d}^{-1}\mathbf{S}'_e\mathbf{d}^{-1}$ . When this is used to calculate the effective moduli of the composite we find

$$\sigma_{\rm e} = e^2 \frac{d_{22}^2 S_{\rm e11}' + d_{12}^2 S_{\rm e22}'}{(d_{11} d_{22} - d_{12}^2)^2} \tag{10}$$

$$\gamma_{e} = k^{2}T \frac{d_{12}^{2}S_{e11}' + d_{11}^{2}S_{e22}'}{(d_{11}d_{22} - d_{12}^{2})^{2}}$$
(11)

$$\alpha_{\rm e} = -\frac{k}{e} \frac{d_{12}}{d_{22}} \left( 1 + \frac{d_{11}d_{22} - d_{12}^2}{d_{12}^2 + d_{22}^2 S_{\rm e11}' / S_{\rm e22}'} \right).$$
(12)

The fact that  $\sigma_e$ ,  $\gamma_e$  and  $\alpha_e$  all depend on the same two quantities  $S'_{e11}$  and  $S'_{e22}$  can be used to derive an exact relation among them [7, 8]:

$$\frac{\alpha_B - \alpha_e}{\alpha_B - \alpha_A} = \frac{\frac{\gamma_e}{\gamma_B} / \frac{\sigma_e}{\sigma_B} - 1}{\frac{\gamma_A}{\gamma_B} / \frac{\sigma_A}{\sigma_B} - 1}.$$
(13)

Note, however, that  $\sigma_e$  depends on all six of the component parameters  $\sigma_A$ ,  $\sigma_B$ ,  $\gamma_A$ ,  $\gamma_B$ ,  $\alpha_A$ ,  $\alpha_B$  and not merely on the pair of electrical conductivities  $\sigma_A$ ,  $\sigma_B$ , as would be the case in the absence of thermoelectricity, when all the  $\alpha$ 's vanish. A similar statement holds for  $\gamma_e$ . The effective figure of merit is now given by [8]

$$Z_{e} = \left(\frac{\gamma_{e}}{\sigma_{e}\alpha_{e}^{2}} - T\right)^{-1} = \frac{1}{T} \frac{\left(\frac{S_{e11}'}{S_{e22}'} + \frac{d_{11}}{d_{22}}\right)^{2}}{\frac{S_{e11}'}{S_{e22}'} \left(\frac{d_{12}}{d_{22}} - \frac{d_{11}}{d_{12}}\right)^{2}}.$$
 (14)

Since  $\alpha_e$  as well as  $Z_e$  depend upon the single ratio  $S'_{e11}/S'_{e22}$ , the critical behaviour is also entirely determined by this ratio. The quantities  $S'_{e11}$ ,  $S'_{e22}$  are the bulk effective

conductivities of two uncoupled conductivity problems in two composites with the same microgeometry but different components. We can, therefore, write

$$\frac{S'_{eii}}{S'_{Mii}} = m\left(\frac{S'_{Iii}}{S'_{Mii}}\right) \qquad \text{for } i = 1, 2 \tag{15}$$

where the detailed form of the function m(h) depends on the precise microgeometry. In the vicinity of the percolation threshold, i.e. where both h and  $\Delta p \equiv p_M - p_c$  are small, this function has a scaling form [9]

$$m(h, \Delta p) = |\Delta p|' \mathscr{F}\left(\frac{h}{|\Delta p|^{t+s}}\right)$$
(16)

where  $\mathcal{F}(z)$  has the following asymptotic forms

$$\mathcal{F}(z) \sim \begin{cases} \text{constant} & \Delta p < 0; \ z \ll 1 & \text{regime I} \\ z & \Delta p < 0; \ z \ll 1 & \text{regime II} \\ z^{(t/(t+s))} & z \gg 1 & \text{regime III.} \end{cases}$$
(17)

These asymptotic forms will suffice for discussing the different asymptotic regimes of the thermoelectric behaviour. However, when one is interested in the thermoelectric behaviour throughout the critical region, one needs the explicit form of the scaling function  $\mathcal{F}(z)$  for all values of z. For that purpose we shall use the parametric form suggested and used earlier by Straley [10]

$$\mathscr{F} = \lambda^{t}(1+x)$$
  $\lambda x = \Delta p$   $z = \lambda^{s+t}(1-x^{2}).$  (18)

Here t and s are the conductivity critical exponents, which in three-dimensional percolation have the universal values  $t \approx 2.0$  [11] and  $s \approx 0.75$  [12].

### 3. Scaling behaviour near percolation

In our study we always assumed that  $\sigma_I / \sigma_M$ ,  $\gamma_I / \gamma_M$  and  $|\Delta p|$  are all small compared to unity. We then fixed the two ratios and allowed  $|\Delta p|$  to vary. The critical behaviour of the bulk effective thermoelectric moduli of the composite was studied numerically, using expressions (12), (14) and (18). The behaviour of  $\alpha_e$  was also studied analytically for the limit of very small Seebeck coefficients in both components, which is the common case in practice, using expressions (13) and (17).

### 3.1. Numerical results

Typical results for the critical behaviour of  $\alpha_e$  are shown in figure 1. We see three different types of behaviour depending on the electrical conductivities ratio and the thermal conductivities ratio of the components. One type of behaviour appears when  $\sigma_I/\sigma_M \ll \gamma_I/\gamma_M \ll 1$  (the common case in practice). Then, if we decrease  $p_M$  starting above  $p_e$ , the effective thermopower increases very slowly at first, staying very close to  $\alpha_M$ . At  $p_e$ , and even slightly below it, it is still nearly equal to the pure good conductor value  $\alpha_M$ . Below  $p_e$  the effective thermopower increases abruptly, levelling off to the bad conductor value  $\alpha_I$  only a considerable distance below  $p_e$ . In the other case  $\gamma_1/\gamma_M \ll \sigma_I/\sigma_M \ll 1$ , which is much less common in practice, a similar sequence of events is found to occur when  $p_M$  increases through  $p_e$  (see figure 1): At first  $\alpha_e$ 



Figure 1. The effective Seebeck coefficient  $\alpha_e$  of a good conductor-bad conductor composite versus the good conductor concentration  $p_M$ , for different values of the thermal and electrical conductivity ratios.

remains very close to  $\alpha_I$ , then an abrupt decrease occurs above  $p_c$ , while levelling out to  $\alpha_M$  takes place only when  $p_M$  is considerably above  $p_c$ . Finally, when the two conductivity ratios have similar magnitudes, the transition at percolation, from  $\alpha_M$  to  $\alpha_{I}$ , is almost symmetric about  $p_{c}$  (see figure 1). These qualitative features do not depend on the precise values of the Seebeck coefficients of the components. However, the detailed shape of the transition in the first case (the common one) does depend on the value of the thermoelectric coefficient of the good conductor, as shown in figure 2: As  $\alpha_M$  increases and approaches its maximum allowed value of  $\sqrt{\gamma_M/(\sigma_M T)}$ , the transition becomes less abrupt. For values of  $\alpha_M$  very close to this limit the transition is so slow that it is obvious it cannot describe the real behaviour outside of a very small region below  $p_c$ . For comparison, we calculated the critical behaviour for these cases using Bruggeman's symmetric effective medium approximation for the uncoupled effective conductivities  $S'_{e11}$  and  $S'_{e22}$  [13]. This calculation should give good results for small volume fractions of the good conductor, far below  $p_c$ . These results are shown in figure 3. Comparing figures 2 and 3 it is apparent that for small  $\alpha_M$  (i.e. far from its upper limit) the scaling calculation gives good results in a large range below  $p_{\rm c}$ , whereas for values of  $\alpha_M$  that are close to the upper limit, the behaviour far below  $p_c$  changes considerably and the result of the scaling approximation is not valid. Such changes in the critical behaviour with variations of either  $\alpha_M$  or  $\alpha_I$  do not arise in the opposite case, when  $\alpha_e$  is dominated by the bad conductor.

The effective thermoelectric figure of merit  $Z_e$  can be similarly calculated. Unlike  $\alpha_e$ ,  $Z_e$  is not always a monotonic function of the ratio  $S'_{e11}/S'_{e22}$ . It has a single minimum when  $S'_{e11}/S'_{e22} = |d_{11}/d_{22}|$ , if that value is inside the range of allowed values [8]. Therefore, if  $|d_{11}/d_{22}|$  lies between the pure component values of  $S'_{e11}/S'_{e22}$ , this minimum is attained when  $p_M$  is varied, otherwise the minimum is not attained and the transition is monotonic. The condition for the attainability of this minimum,



Figure 2. The effective Seebeck coefficient  $\alpha_e$  for several values of  $\alpha_M$ , up to the upper limit  $\alpha_M = 8.513 \times 10^{-5}$ , versus the good conductor concentration  $p_M$ .



Figure 3. The same as figure 2, but using the effective medium approximation.

expressed in terms of the properties of the components is [14]:

$$\frac{\frac{\gamma_M}{\sigma_M}\alpha_I - \frac{\gamma_I}{\sigma_I}\alpha_M}{\alpha_M - \alpha_I} > 0.$$
(19)

Results of the scaling calculation of  $Z_e$  for the same cases as shown in figure 1 are shown in figure 4. The behaviour is found, again, to depend on the relation between



Figure 4. The effective figure of merit  $Z_e$  of a good conductor-bad conductor composite versus the good conductor concentration  $p_M$ , for the same cases as figure 1, using the scaling function of equation (18).

the thermal conductivities ratio  $\gamma_I/\gamma_M$  and the electrical conductivities ratio  $\sigma_I/\sigma_M$ . When the two ratios are similar the transition at percolation, from  $Z_M$  to  $Z_I$ , is almost symmetric about  $p_c$ . If  $\gamma_I/\gamma_M \ll \sigma_I/\sigma_M \ll 1$ , then  $Z_e$  still changes monotonically with  $\Delta p$  from the good conductor value  $Z_M$  to the bad conductor value  $Z_I$ , but the transition is no longer symmetric about  $p_c$ . At the threshold,  $Z_e$  is much smaller than in the first case. In the other case  $\sigma_I/\sigma_M \ll \gamma_I/\gamma_M \ll 1$ , and when the condition (19) is also satisfied (this is the common case in practice),  $Z_e$  is non-monotonic and has a minimum slightly above  $p_c$ . The location of the minimum is determined by the pure component values of Z. As  $Z_M$  increases the minimum moves to smaller values of  $p_M$ , and may even cross to the other side of the percolation threshold  $p_c$ . A graph similar to our figures 1 and 4 for a particular case of the common type only was calculated and plotted by Straley [5].

The same calculations can be made starting from (13) and using (18) for the effective conductivities  $\sigma_e$  and  $\gamma_e$ . However, the results of such a calculation will be less accurate than those of the previous one, since  $\sigma_e$  and  $\gamma_e$  are not the uncoupled conductivities for which (18) is strictly valid. In order for this procedure to be a good approximation, the off-diagonal elements of the matrix S of (3) must be small compared to the diagonal elements. This is usually true in practice for each component and, consequently, also for the composite. The results of this calculation are then very similar to those of the previous, more exact one. The behaviour of the effective thermoelectric properties exhibits the same general features in each of the cases described above.

## 3.2. Analytic results

In order to analyse the different possible modes of behaviour analytically, we shall assume the common situation where  $S_{12} \ll S_{11}$  and  $S_{12} \ll S_{22}$  in both components. In that case  $\gamma_e/\gamma_M$  depends only on  $\gamma_I/\gamma_M$  while  $\sigma_e/\sigma_M$  depends only on  $\sigma_I/\sigma_M$ . Under

these conditions we can use the asymptotic scaling forms of (17) directly for the effective conductivities  $\sigma_e$  and  $\gamma_e$ , and then calculate  $\alpha_e$  using (13).

The critical behaviour of the thermoelectric coefficient turns out to be quite rich since it depends on three small dimensionless parameters  $\sigma_I/\sigma_M$ ,  $\gamma_I/\gamma_M$  and  $\Delta p$ , instead of just two parameters as in the familiar case of simple conductivity. This behaviour is again found to depend on the relation between the electrical and thermal conductivity ratios.

In the case that they are similar we find, above  $p_c$ 

$$\frac{\alpha_M - \alpha_e}{\alpha_M - \alpha_I} \propto \left(\frac{\sigma_I}{\sigma_M}\right) (\Delta p)^{-t-s} \qquad \text{for } \frac{\sigma_I}{\sigma_M} \ll (\Delta p)^{t+s} \tag{20}$$

while below  $p_c$  we find  $\alpha_c \approx \alpha_l$ . In order to see the deviations from this value below  $p_c$ , we must consider the second term in the power series expansion of the scaling function  $\mathscr{F}(z) = z + z^2 + \ldots$  (Here and in subsequent expressions we ignore coefficients of the power series—they are expected to be of order 1.) We thus find

$$\frac{\alpha_M - \alpha_e}{\alpha_M - \alpha_I} \simeq 1 - \left(\frac{\sigma_I}{\sigma_M}\right) |\Delta p|^{-t-s} \qquad \text{for } \frac{\sigma_I}{\sigma_M} \ll (\Delta p)^{t+s}.$$
(21)

At  $p_c$ , or very near to it, we find

$$\frac{\alpha_M - \alpha_e}{\alpha_M - \alpha_I} \propto \left(\frac{\gamma_I \sigma_M}{\sigma_I \gamma_M}\right)^{-s/(t+s)} \qquad \text{for } \frac{\sigma_I}{\sigma_M} \gg (\Delta p)^{t+s}.$$
(22)

The behaviour in the two extreme cases is as follows:

 $\begin{array}{l} (\mathscr{A}) \quad \gamma_{I}/\gamma_{M} \gg \sigma_{I}/\sigma_{M} \\ (a) \quad \text{For } p_{M} > p_{c} \text{ and } (\Delta p)^{t+s} > (\gamma_{I}/\gamma_{M}) \text{ we get, as in the previous case,} \\ \quad \frac{\sigma_{M} - \alpha_{c}}{\alpha_{M} - \alpha_{c}} \propto \left(\frac{\sigma_{I}}{\alpha_{M}}\right) (\Delta p)^{-t-s} \ll 1.$  (23)

This means that  $\alpha_e$  remains very close to  $\alpha_M$  as  $\Delta p$  decreases.

(b) For  $p_M > p_c$  in the transition region  $(\sigma_I/\sigma_M) \ll (\Delta p)^{t+s} \ll (\gamma_I/\gamma_M)$  we find

$$\frac{\alpha_M - \alpha_e}{\alpha_M - \alpha_I} \propto \left(\frac{\gamma_M}{\gamma_I}\right)^{s/(t+s)} \left(\frac{\sigma_I}{\sigma_M}\right) (\Delta p)^{-t} \ll 1.$$
(24)

The slow increase continues, but with a different exponent t. The effective thermopower  $\alpha_c$  is still nearly equal to  $\alpha_M$ .

(c) At  $p_c$  we have

$$\frac{\alpha_M - \alpha_e}{\alpha_M - \alpha_I} \propto \left(\frac{\gamma_I \sigma_M}{\gamma_M \sigma_I}\right)^{-s/(t+s)} \ll 1.$$
(25)

This means that  $\alpha_e$  is nearly equal to  $\alpha_M$  even at threshold. It gets smaller with increasing contrast between the thermal conductivities ratio and the electric conductivities ratio.

(d) Below 
$$p_c$$
, for  $(\sigma_I/\sigma_M) \ll |\Delta p|^{t+s} \ll (\gamma_I/\gamma_M)$ , we have  

$$\frac{\alpha_M - \alpha_e}{\alpha_M - \alpha_I} \propto \left(\frac{\gamma_M}{\gamma_I}\right)^{s/(t+s)} |\Delta p|^s - \frac{\gamma_M \sigma_I}{\gamma_I \sigma_M}.$$
(26)

The slow increase continues with the exponent s. The right-hand side is still much smaller than unity so  $\alpha_e$  is dominated by the good conductor value even below  $p_c$ .

(e) Further below  $p_c$ , when  $|\Delta p|^{t+s} > (\gamma_I / \gamma_M)$ , we find

$$\frac{\alpha_M - \alpha_e}{\alpha_M - \alpha_I} \simeq 1 - \frac{\sigma_I}{\sigma_M} |\Delta p|^{-t-s}.$$
(27)

Now the RHS is very close to unity so  $\alpha_e$  is only slightly smaller than  $\alpha_I$ . The transition between (d) and (e) is very sharp and cannot be discerned from this type of simple scaling approach.

(B) 
$$\gamma_I / \gamma_M \ll \sigma_I / \sigma_M$$
  
(a) For  $p_M > p_c$  and  $(\Delta p)^{i+s} > (\sigma_I / \sigma_M)$  we get, as before,

$$\frac{\alpha_M - \alpha_e}{\alpha_M - \alpha_I} \propto \left(\frac{\sigma_I}{\sigma_M}\right) (\Delta p)^{-i-s} \ll 1$$
(28)

so that  $\alpha_e$  is only slightly greater than  $\alpha_M$ .

(b) Above  $p_c$ , for  $(\gamma_I/\gamma_M) \ll (\Delta p)^{t+s} \ll (\sigma_I/\sigma_M)$ , we find

$$\frac{\alpha_M - \alpha_e}{\alpha_M - \alpha_I} \simeq 1 - \left(\frac{\sigma_M}{\sigma_I}\right)^{t/(t+s)} (\Delta p)^t.$$
<sup>(29)</sup>

The RHS is almost unity so  $\alpha_e$  is now close to its bad conductor value  $\alpha_I$  already above  $p_e$ . Here the sharp transition occurs above  $p_c$ .

(c) At  $p_c$ ,  $\alpha_e$  is nearly equal to  $\alpha_I$ . It gets closer to  $\alpha_I$  with increasing contrast between  $\gamma_I/\gamma_M$  and  $\sigma_I/\sigma_M$ .

d) For 
$$p_M < p_c$$
 and  $(\gamma_I / \gamma_M) \ll (\Delta p)^{t+s} \ll (\sigma_I / \sigma_M)$ , we get  

$$\frac{\sigma_M - \alpha_c}{\alpha_M - \alpha_I} \approx 1 - \left(\frac{\gamma_I}{\gamma_M}\right) \left(\frac{\sigma_I}{\sigma_M}\right)^{-t/(t+s)} |\Delta p|^{-s}$$
(30)

so that  $\alpha_e$  continues to rise towards  $\alpha_I$  with the exponent s.

(e) For below  $p_c$ , when  $|\Delta p|^{t+s} > (\sigma_I / \sigma_M)$ , we get  $\alpha_e = \alpha_I$ .

These results agree well with the results of the numerical calculations described before. Analytic scaling calculations for  $Z_e$  produce less clear-cut results, although a considerable qualitative difference in behaviour between the extreme cases is apparent.

#### Acknowledgments

This research was supported, in part, by a grant from the US-Israel Binational Science Foundation.

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